

Modes of Strand Damage Observed by Magneto-Optical Imaging and Metallography

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ASCI08



The Applied Superconductivity Center
The National High Magnetic Field Laboratory
Florida State University



Motivation

- Many factors degrade the performance of Nb_3Sn strands:
 - Plastic deformation during cabling
 - Lorentz force during operation at high field
 - Precompression during magnet fabrication
- In this study, the impact of cabling deformation on Nb_3Sn strand microstructure and properties under controlled conditions has been simulated using rolling (performed at Fermilab).

Summary of Microstructural and Property Measurements Reported Here

Microstructure

-  Scanning Laser Confocal Microscopy (SLCM)
-  Field emission SEM

Spatially Resolved Superconducting Properties

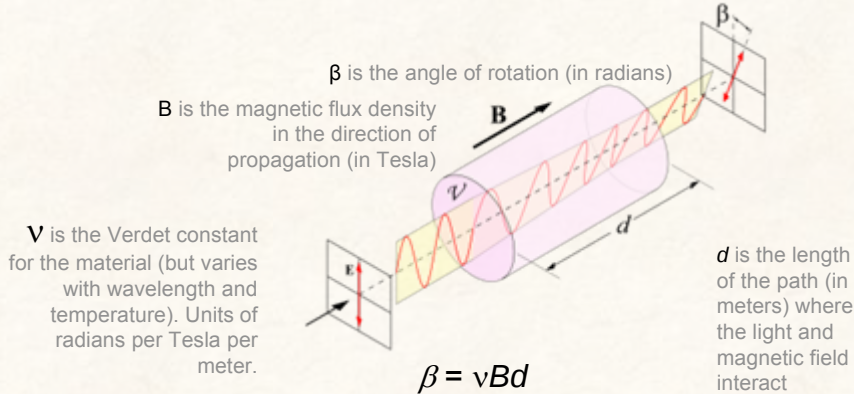
-  Magneto-Optical Imaging (MOI)

Electromagnetic measurements:

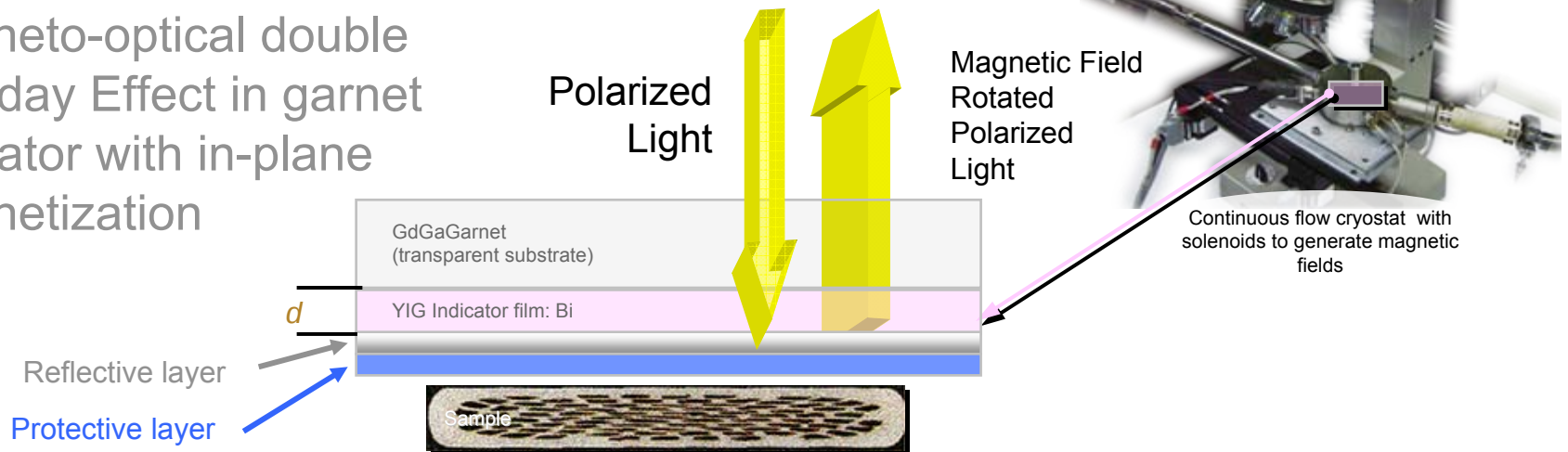
-  I_c , T_c and magnetization measurements

Principle Of Magneto-optical Imaging

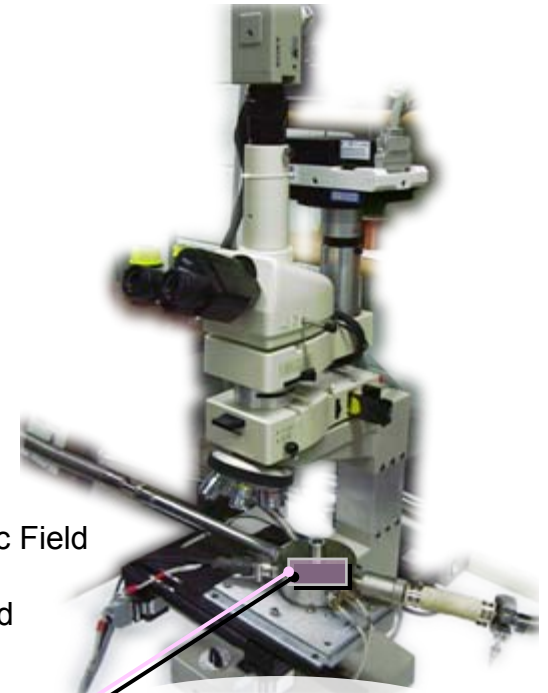
Faraday Effect: The rotation of the plane of polarization is proportional to the intensity of the component of the magnetic field in the direction of the beam of light.



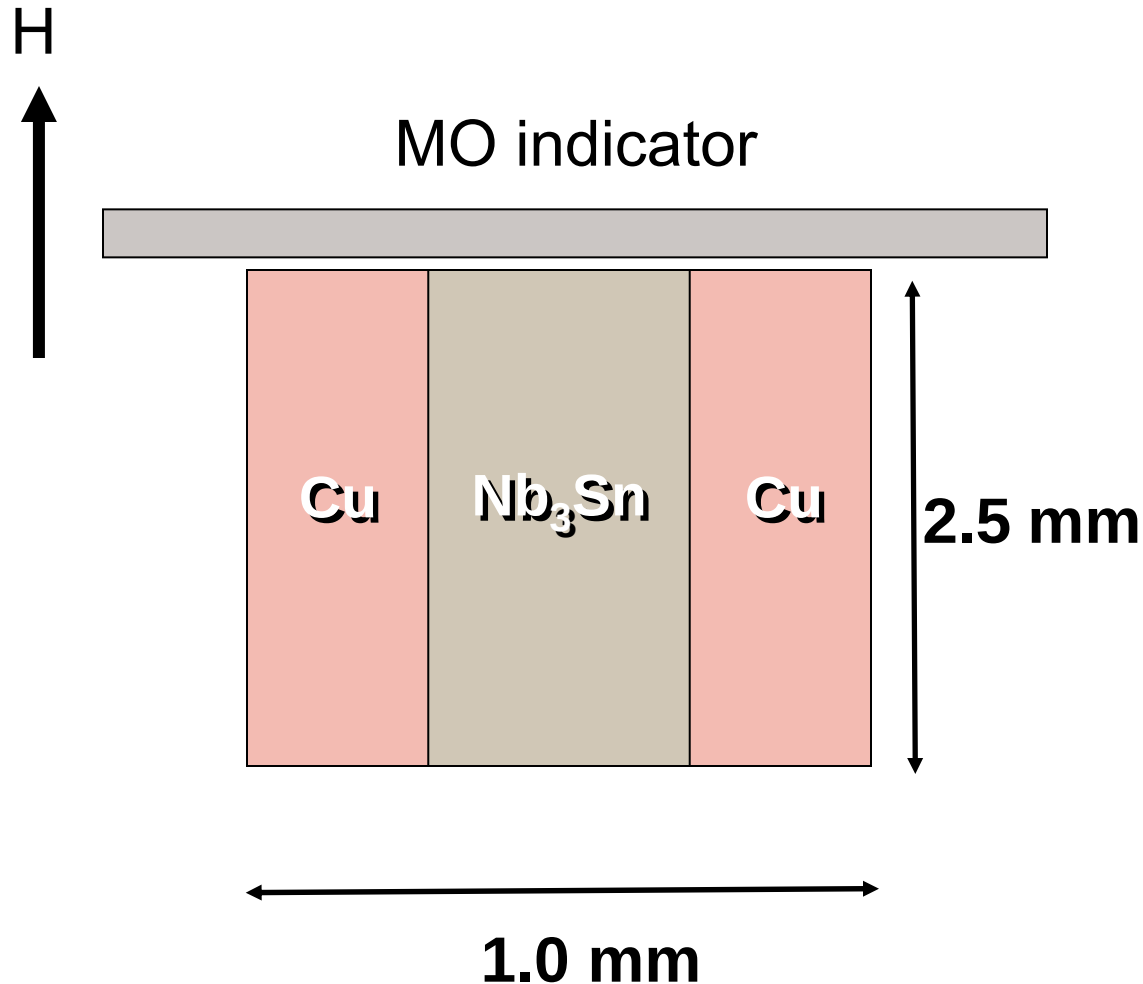
Magneto-optical double Faraday Effect in garnet indicator with in-plane magnetization



Polarized light microscope



Magneto-Optical Experiment on Nb₃Sn Strand



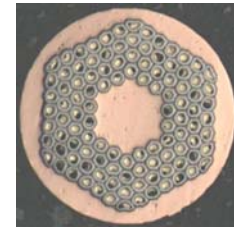
SAMPLES

✚ Strands manufactured by:

✚ Oxford Superconducting Technology (OST)

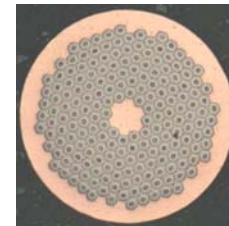
✚ IT- Internal Tin strands

RRP - Restack Rod Process,



✚ ShapeMetal Innovation (SMI),

PIT - Powder-In-Tube

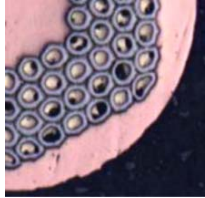


✚ Sample Conditions Examined

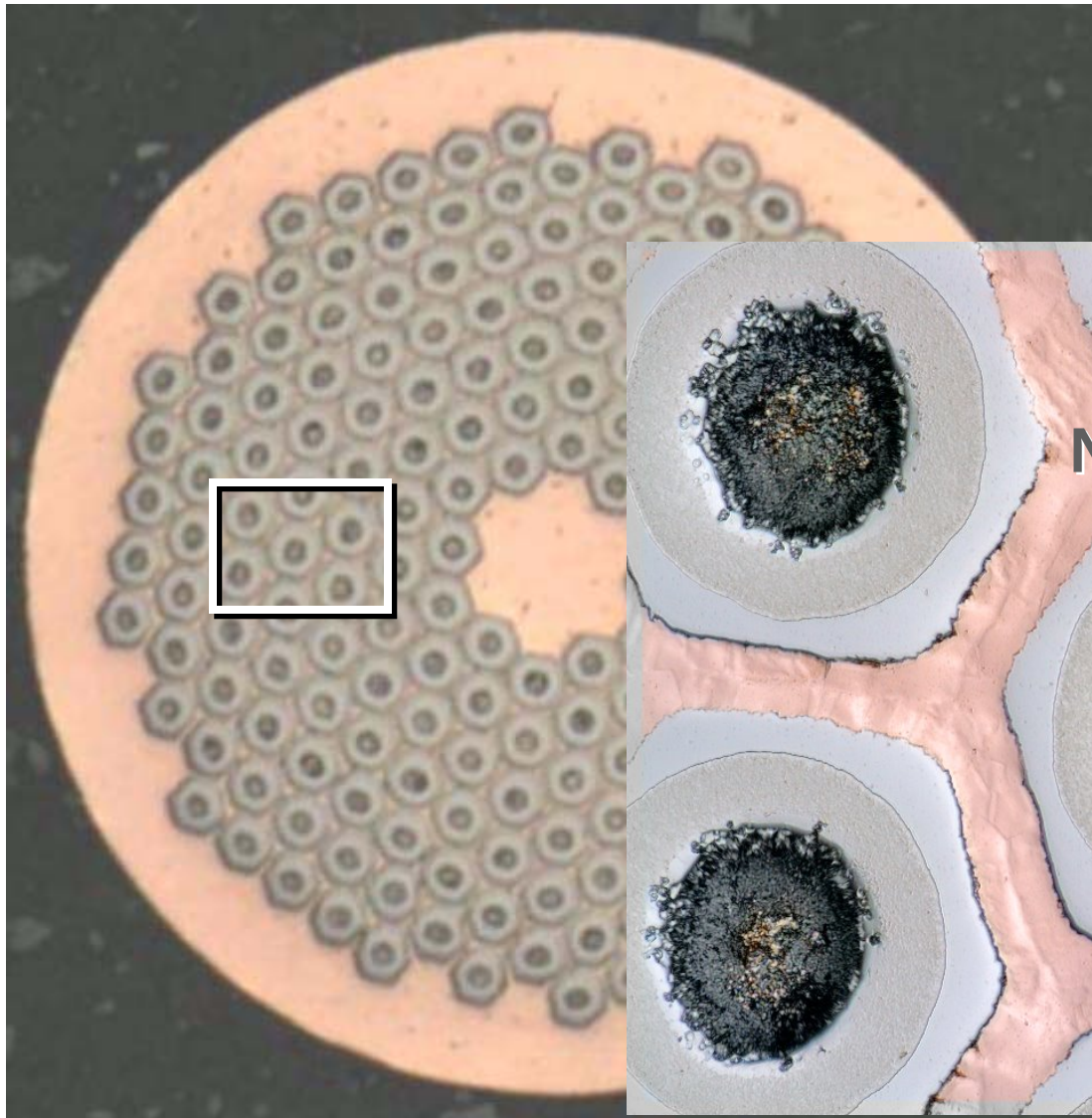
✚ Virgin (round 1 mm) strands

✚ Strands Flat-Rolled down a maximum reduction of 50% strand thickness.

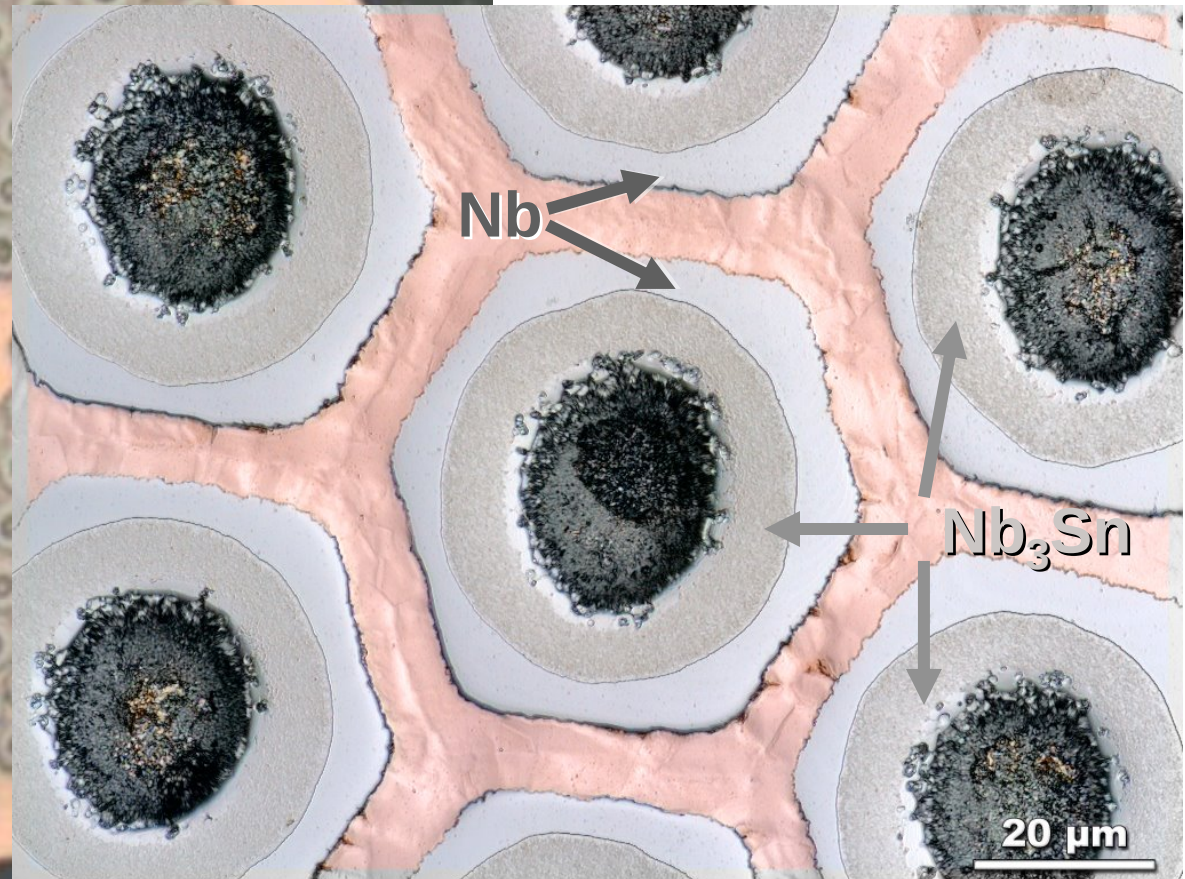
SAMPLE DESCRIPTION

Strand ID	PIT 	RRP 
Billet ID	187	8195-97
No. of filaments	192	108/127
Strand diameter, mm	1.0	1.0
Alloying Element	Ta	Ti (Nb-47Ti rods)
I _c (12T, A)	~700	~900
Deff, μm	45	84
Geometric filament size, μm	50-57	64-75
RRR	250	300
Twist pitch, mm	20	12
Cu fr., %	52	49

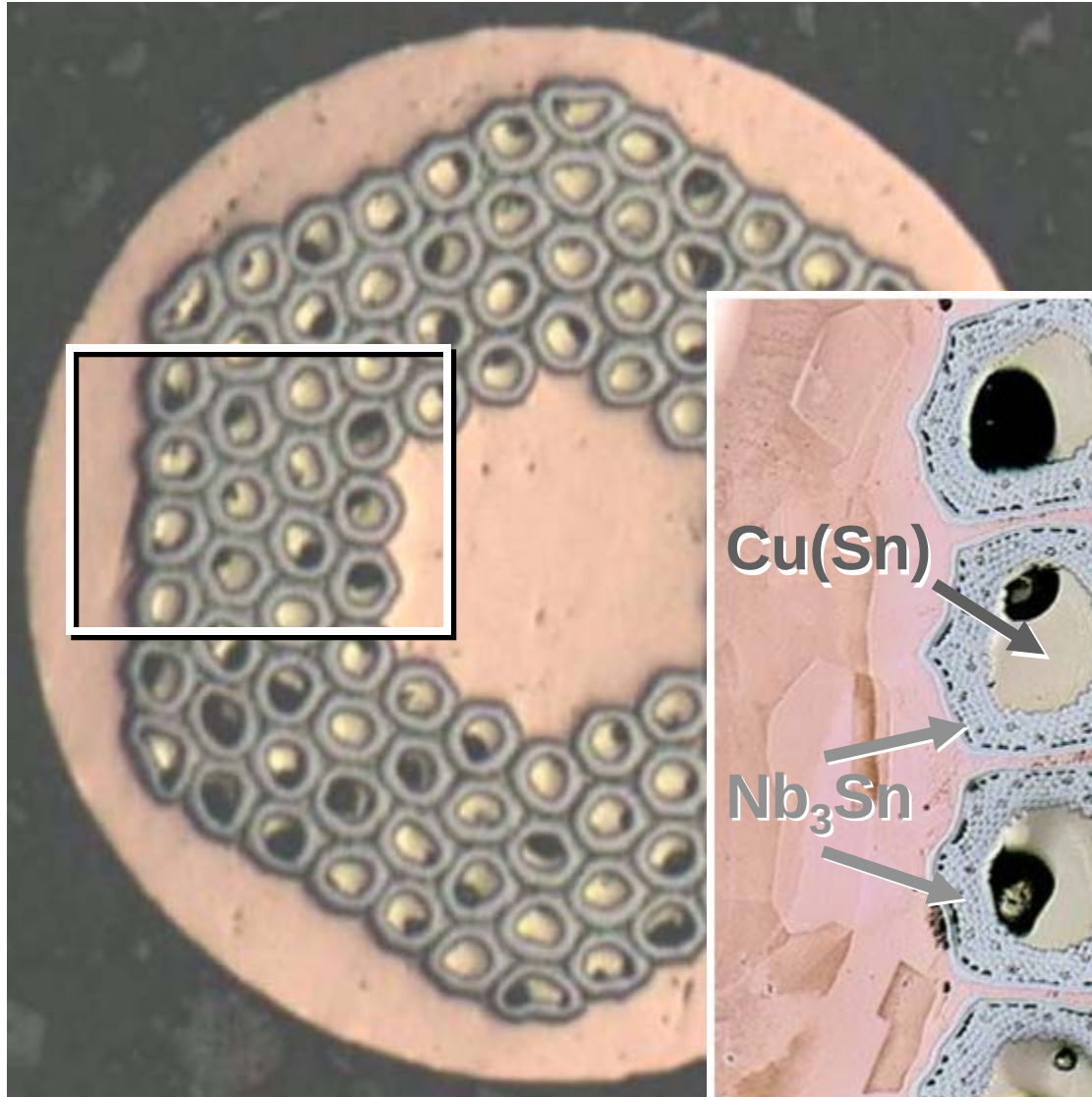
Cross-section of Virgin 1 mm Diam. PIT



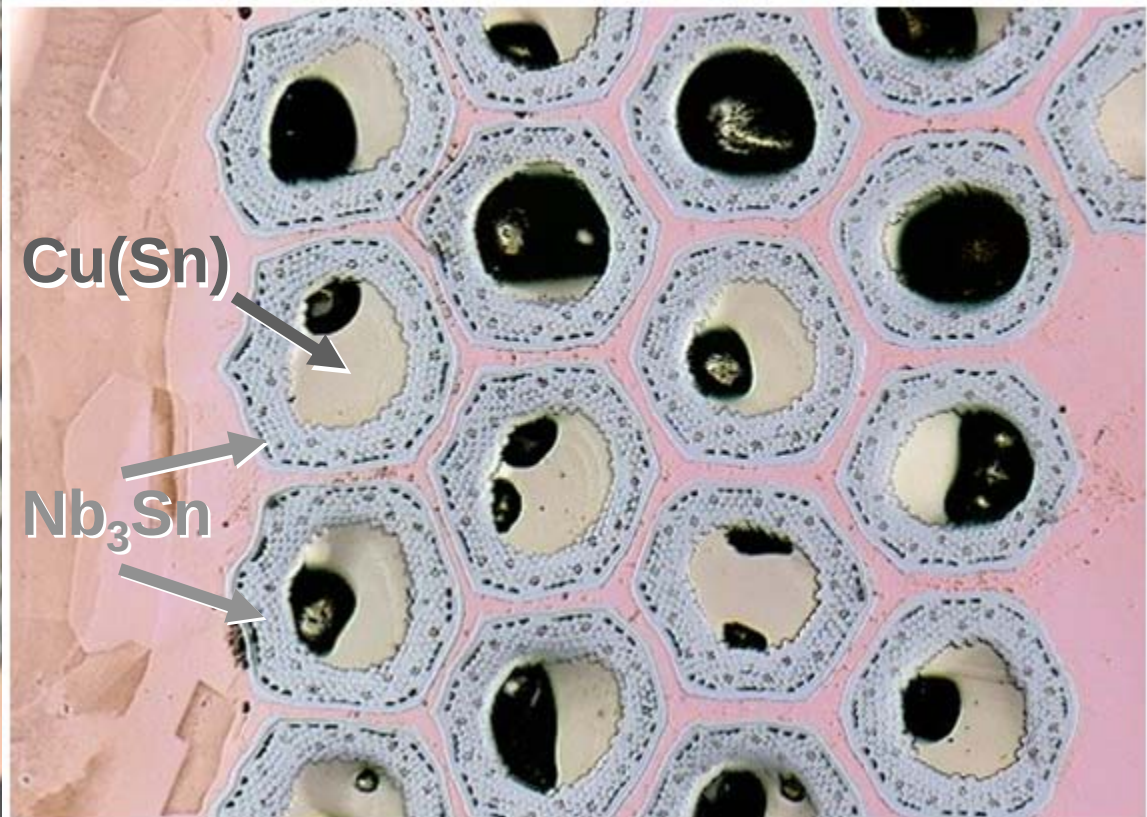
After chemical etching



Cross-section of Virgin 1 mm Diam. IT (RRP)

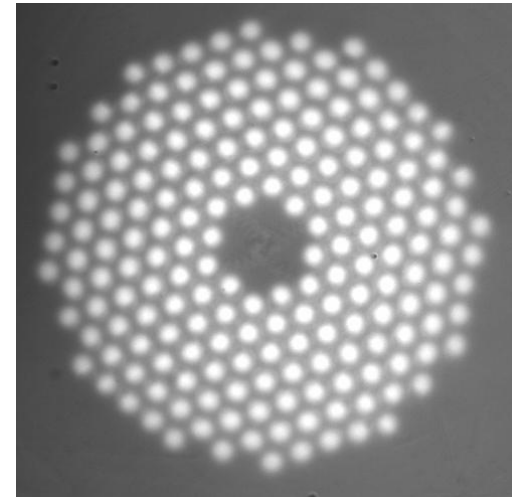
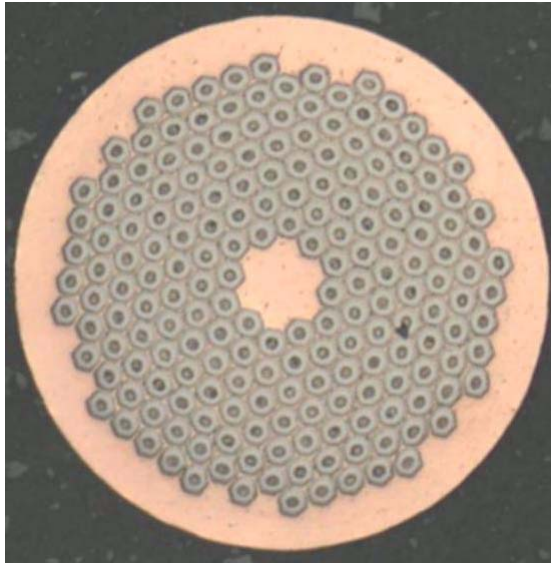


After chemical
etching



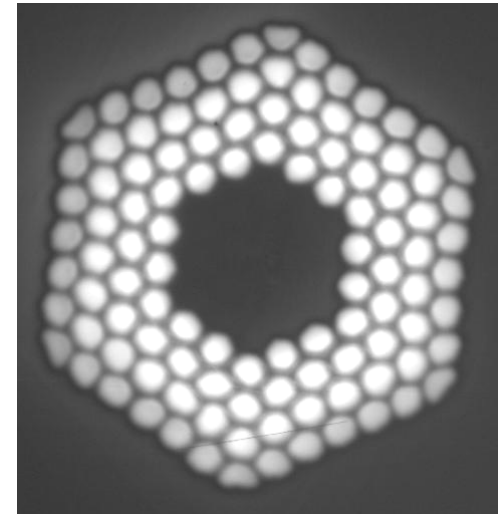
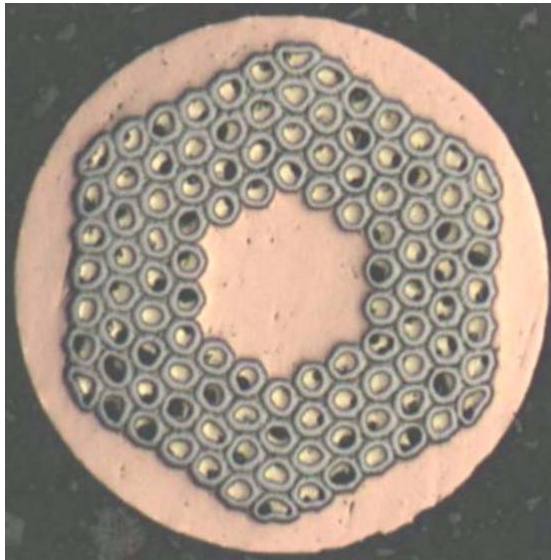
MOI at $T=12$ K and $H=0$ (after FC in $H=120$ mT)

PIT



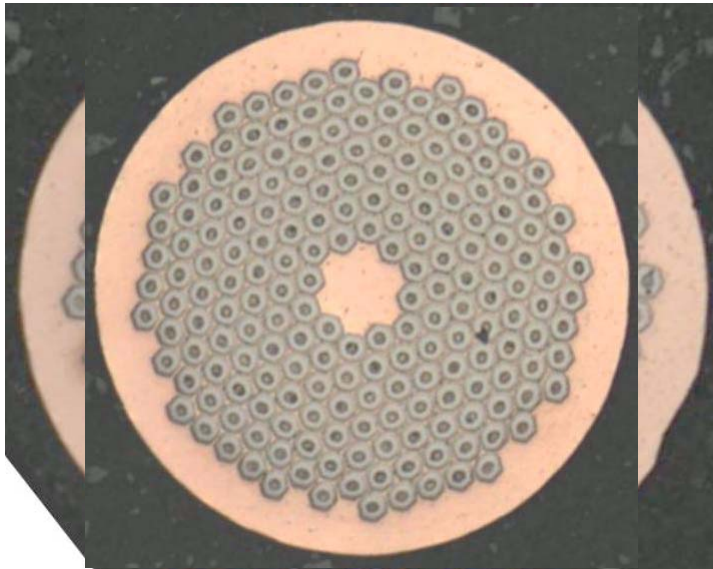
MO images of trapped flux

RRP

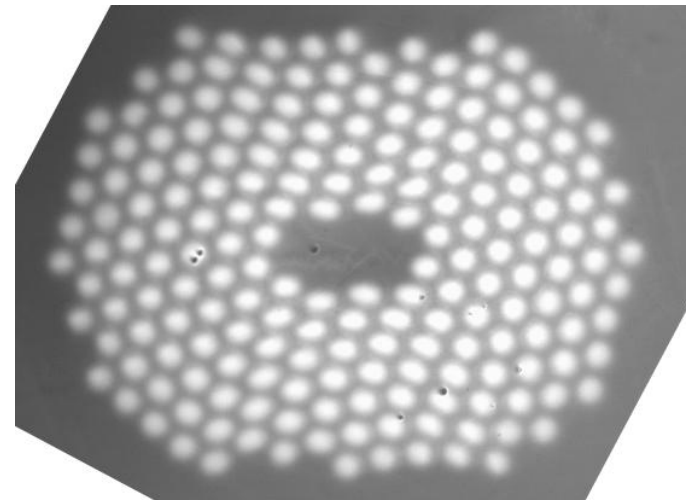


PIT ROLLED TO 0.8 mm : OPTICAL AND MO IMAGES at T=12K, H=0

H=0, FC in H=120 mT

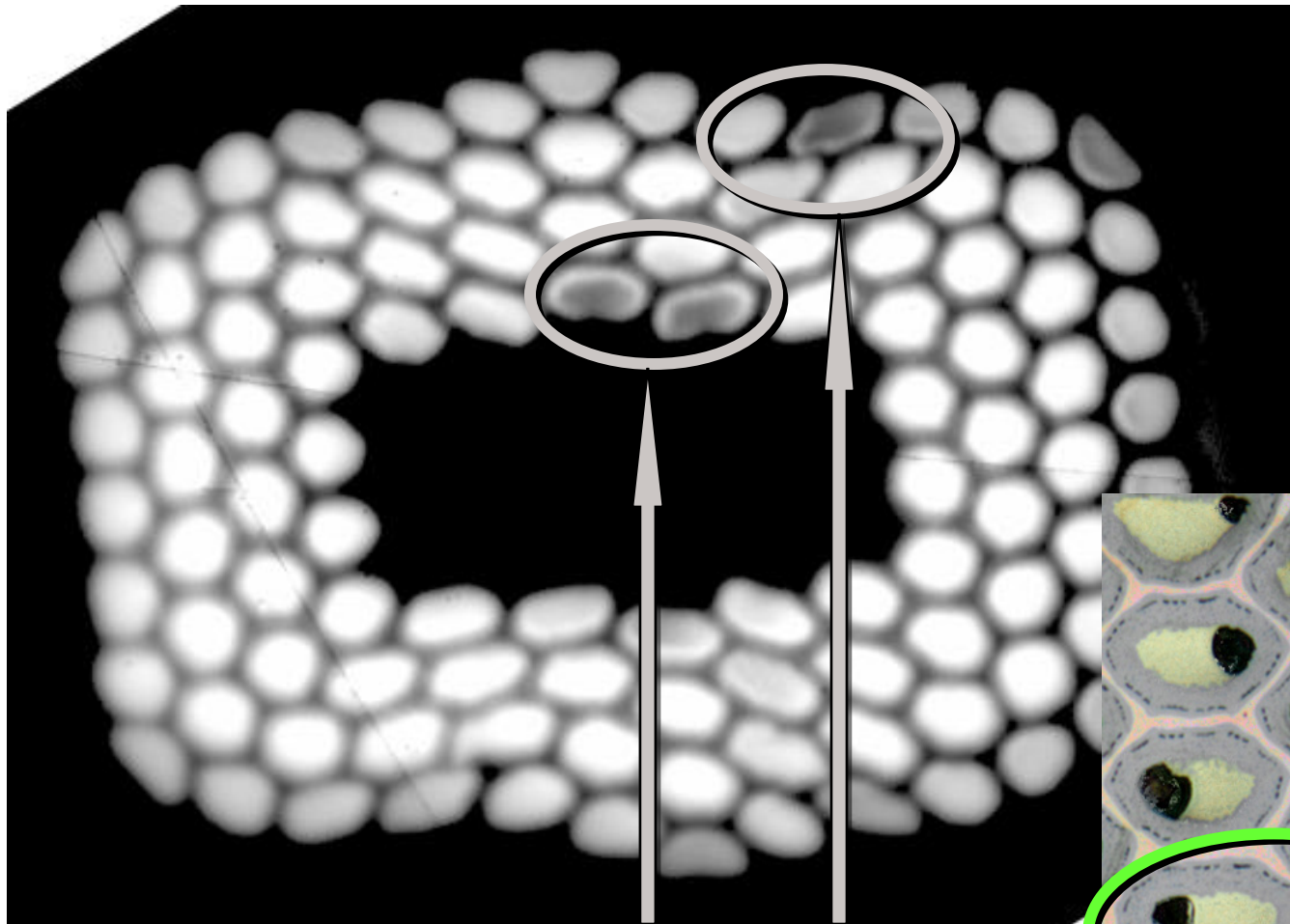


No damage to filament integrity observed by microscopy filament

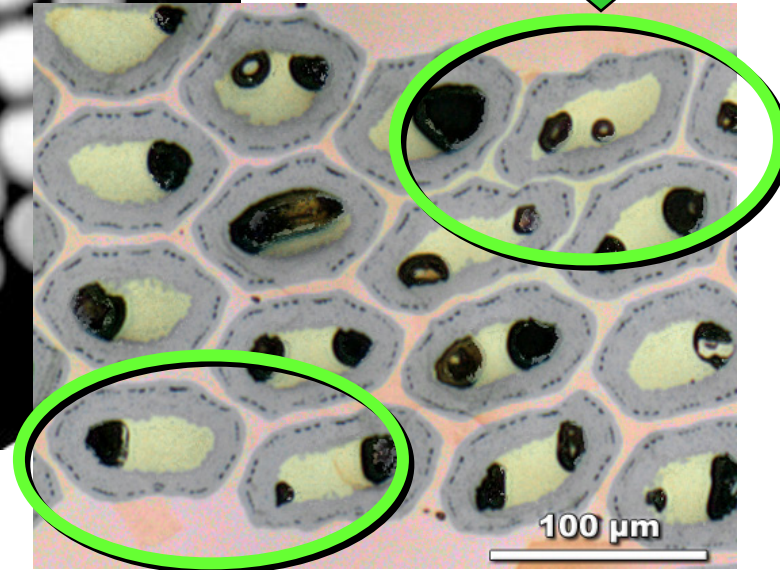


MO shows no significant variation in properties from filament to filament

IT RRP Rolled to 0.8 mm: MO T=12 K, H=0

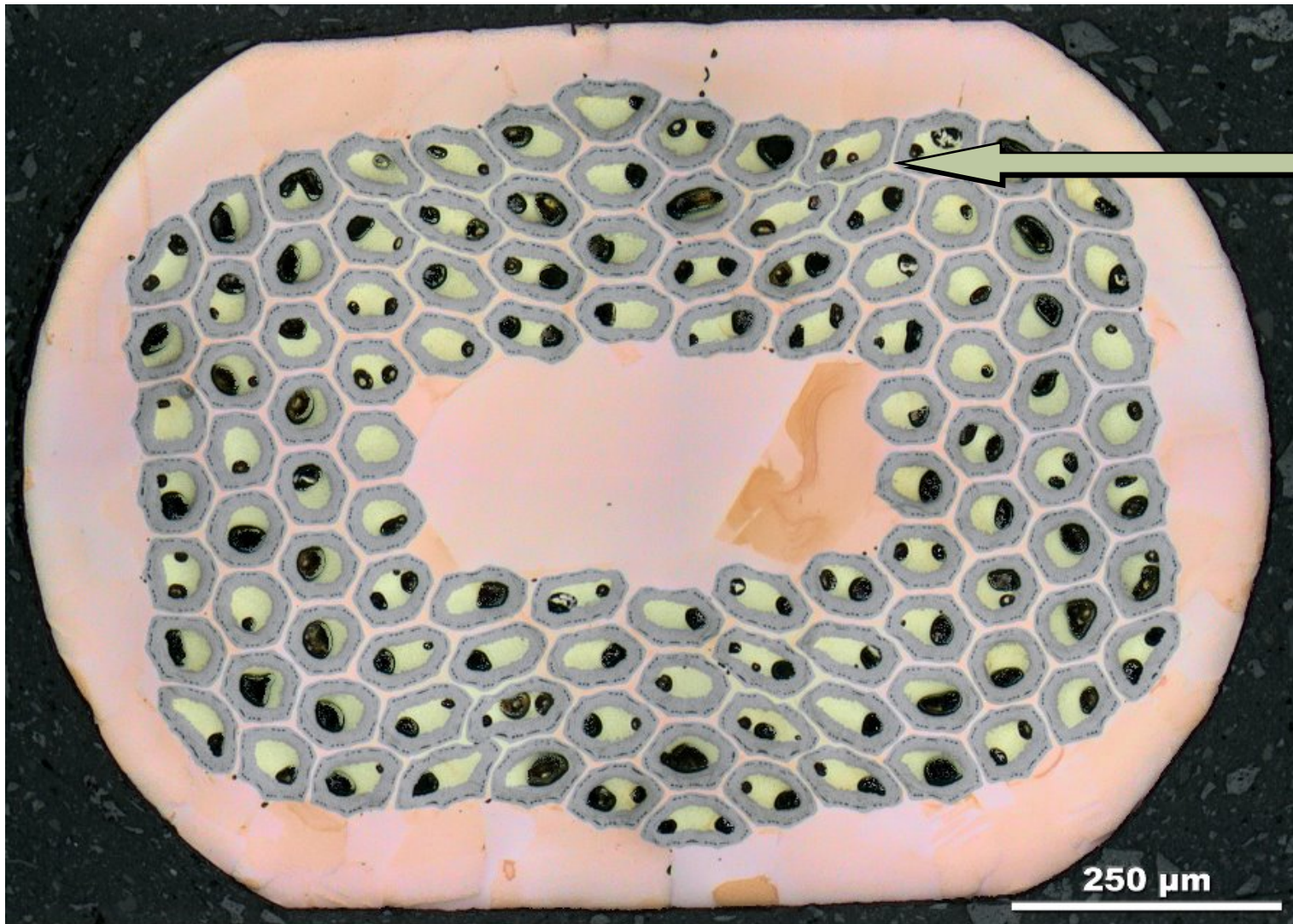


LSCM+color
image shows Sn
leakage between
outer sub-
elements only



*MO shows damage to both
outer and inner sub-elements*

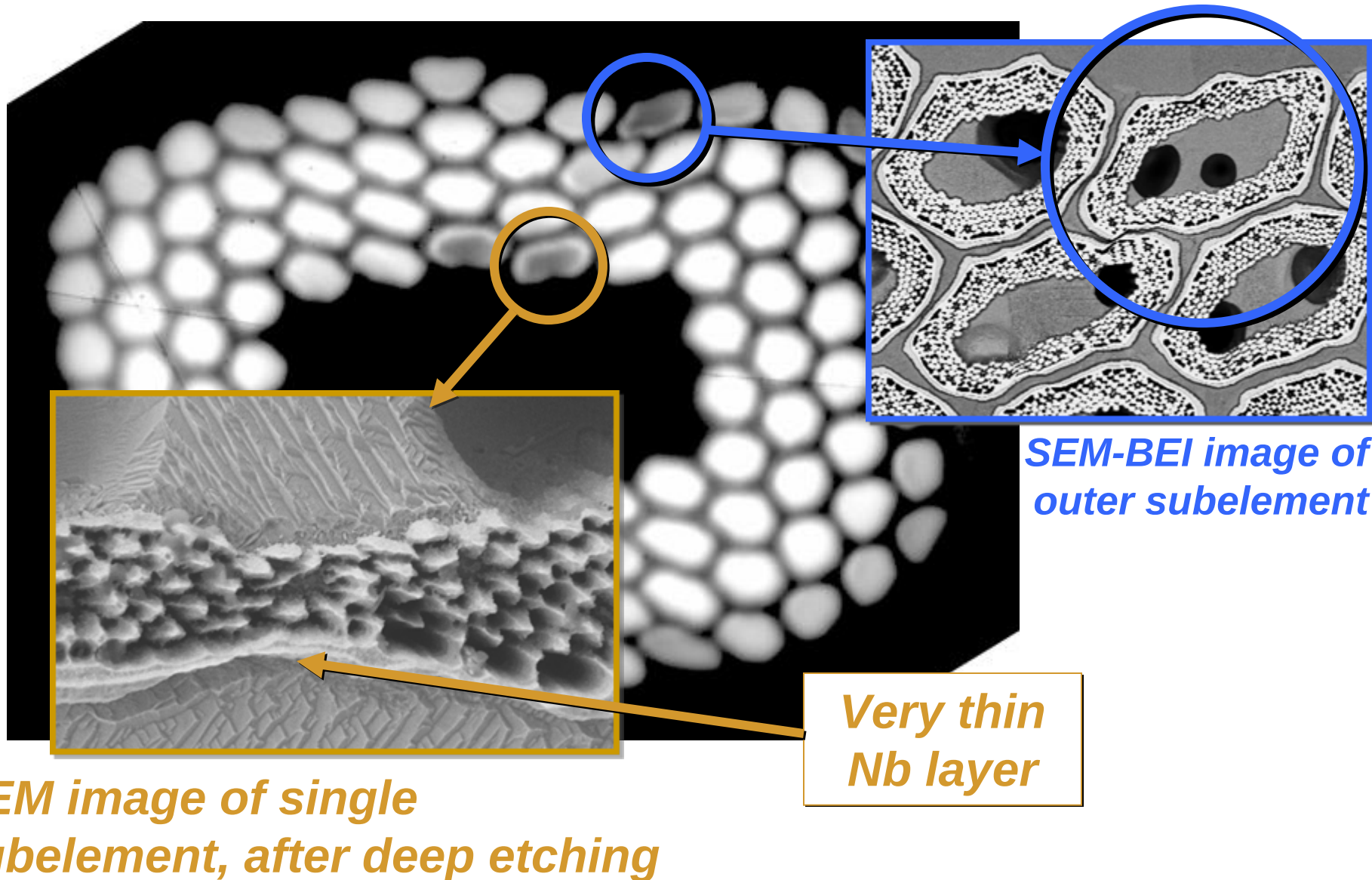
IT RRP Rolled to 0.8 mm : Cross-Section



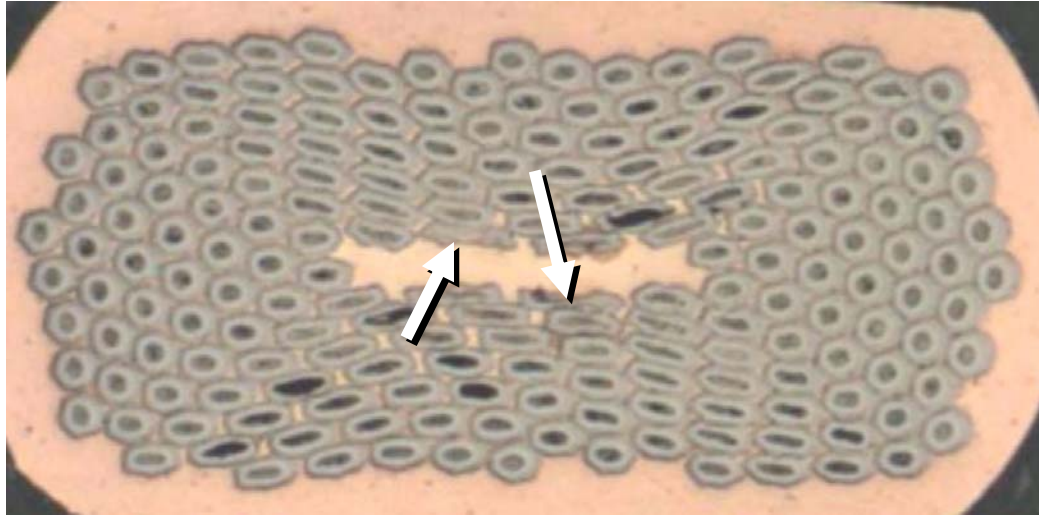
Sub-elements not broken but some Sn leakage observed by color contrast

LSCM/LED-confocal image with Sn-rich Cu enhanced with green tint

IT RRP Rolled to 0.8 mm: SEM of Sub-Element

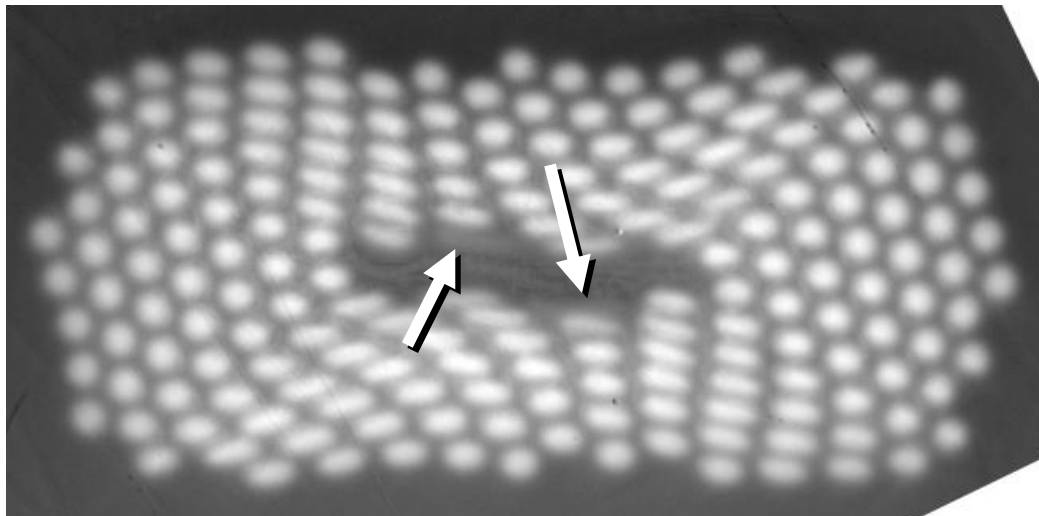


PIT Rolled to 0.6 mm: Color & MO: T=12 K, H=0



Only a few central filaments have been broken by the rolling.

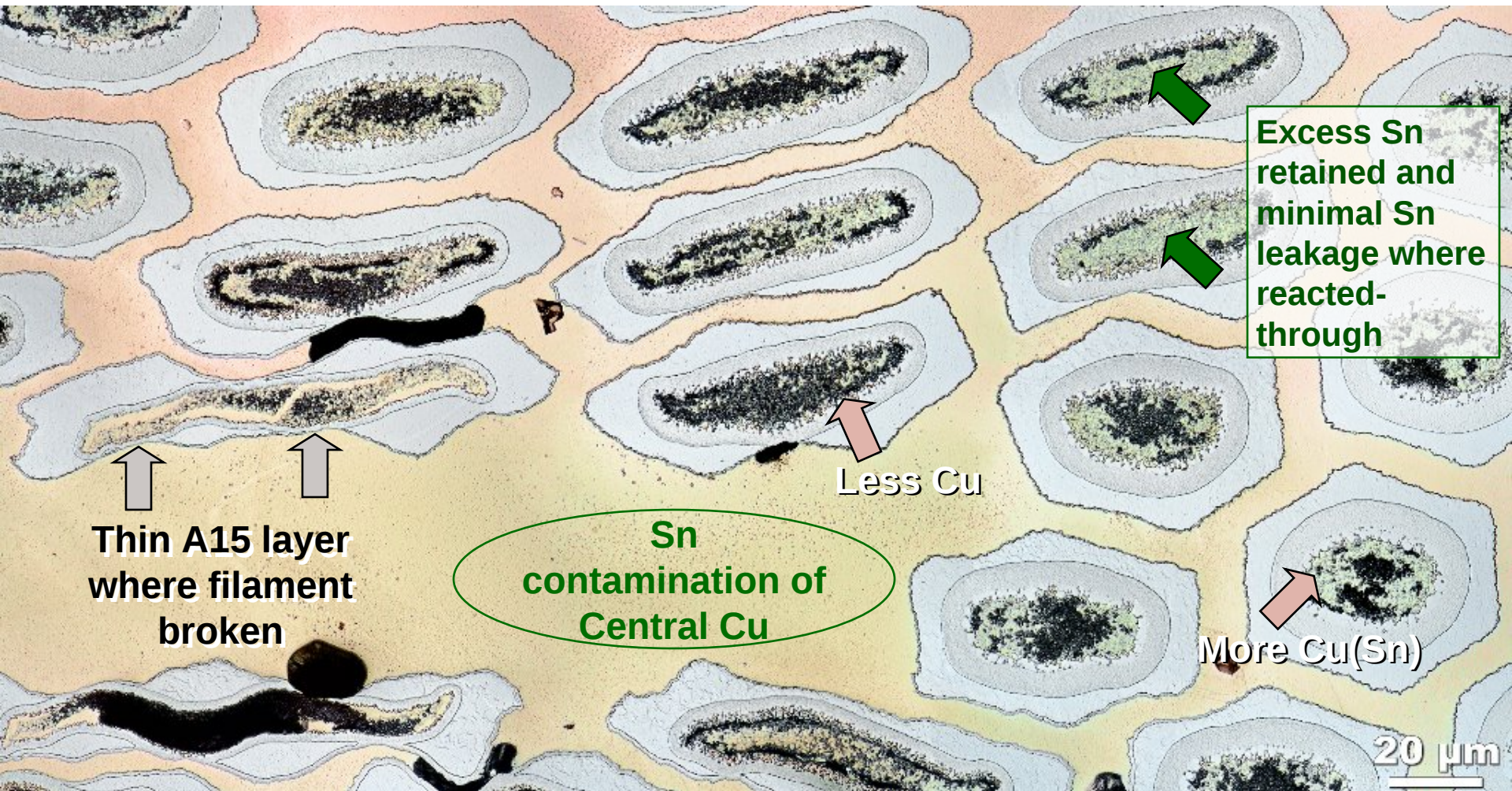
But all the superconductivity in those filaments is lost at 12 K



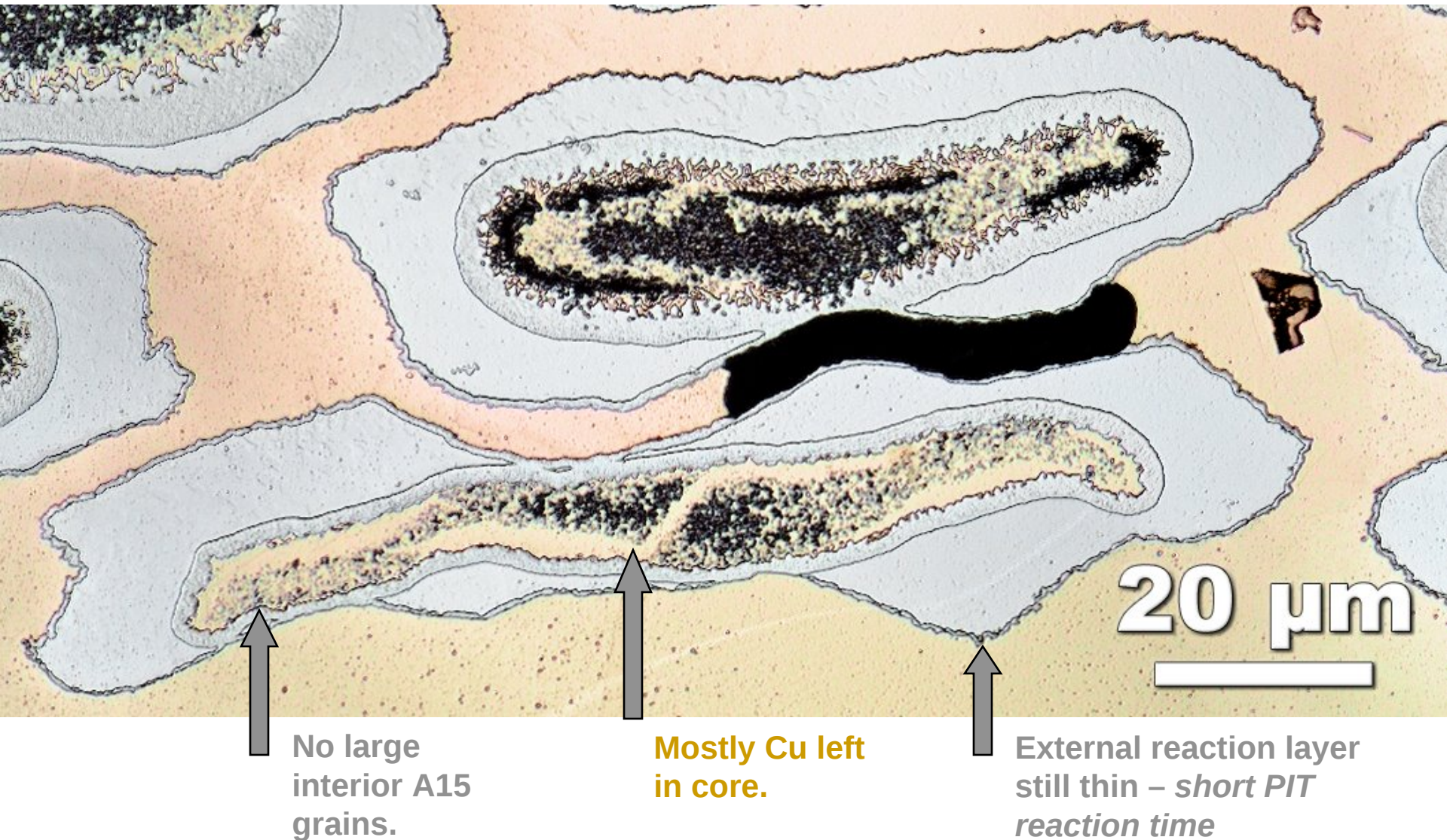
When PIT filaments break . . .

PIT Rolled to 0.6 mm

LSCM+color enhanced to shows Sn leakage

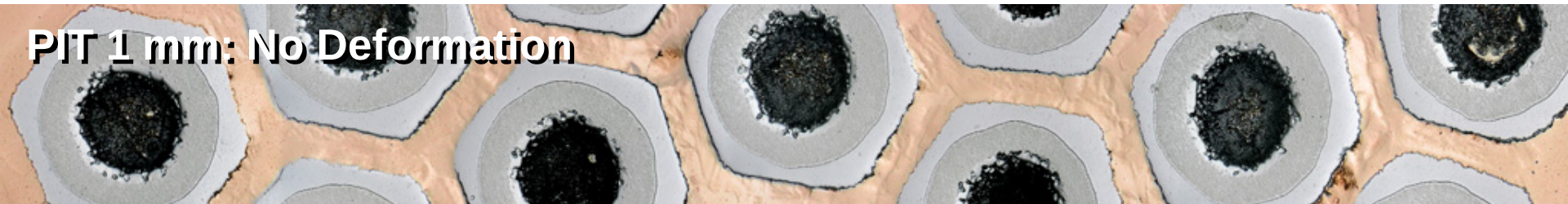


Detail from PIT Rolled to 0.6 mm

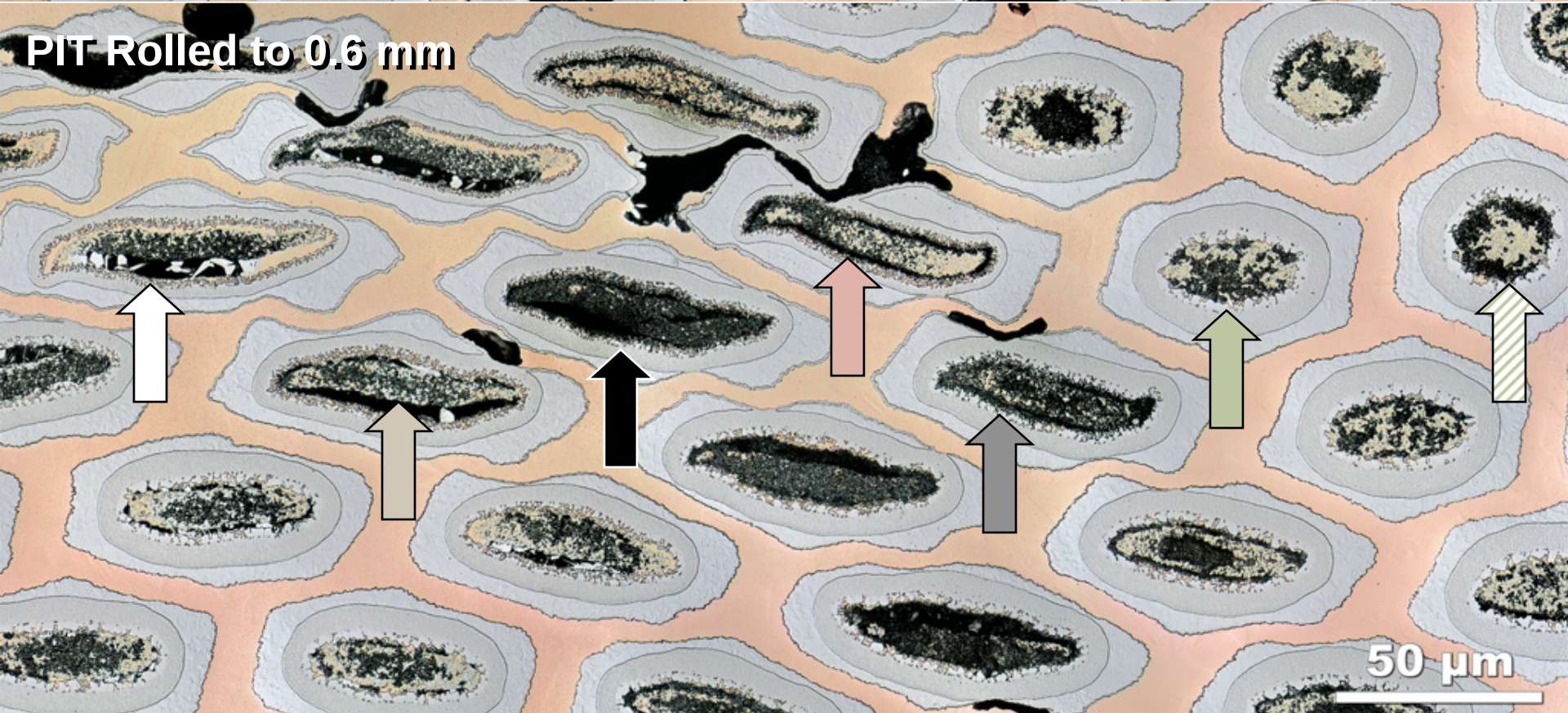


PIT Core Components Change After Rolling

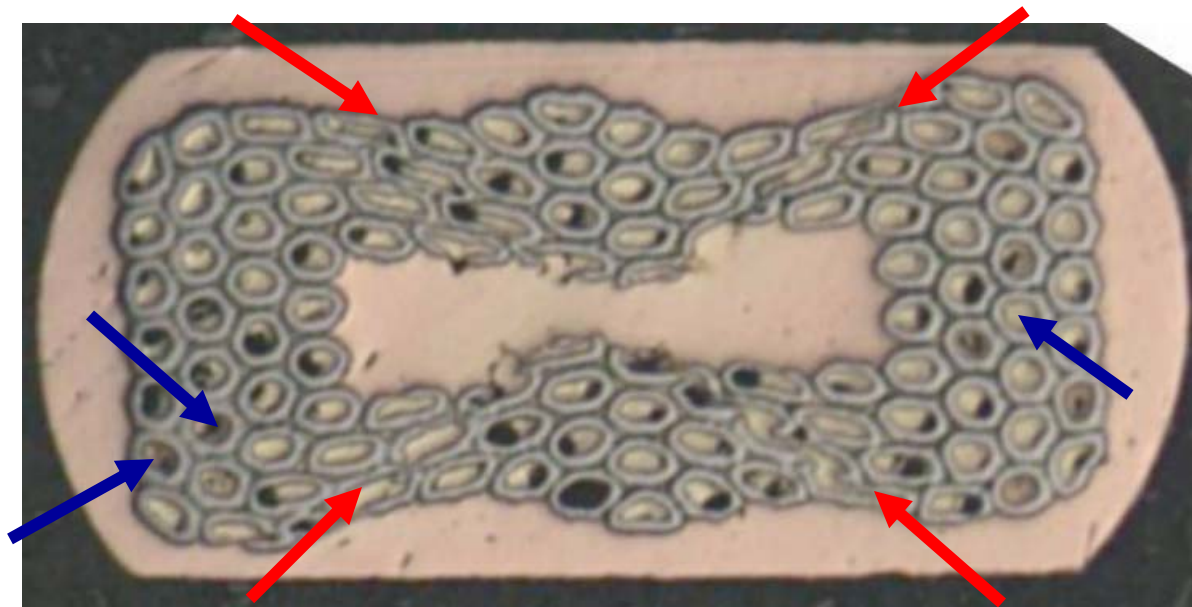
PIT 1 mm: No Deformation



PIT Rolled to 0.6 mm



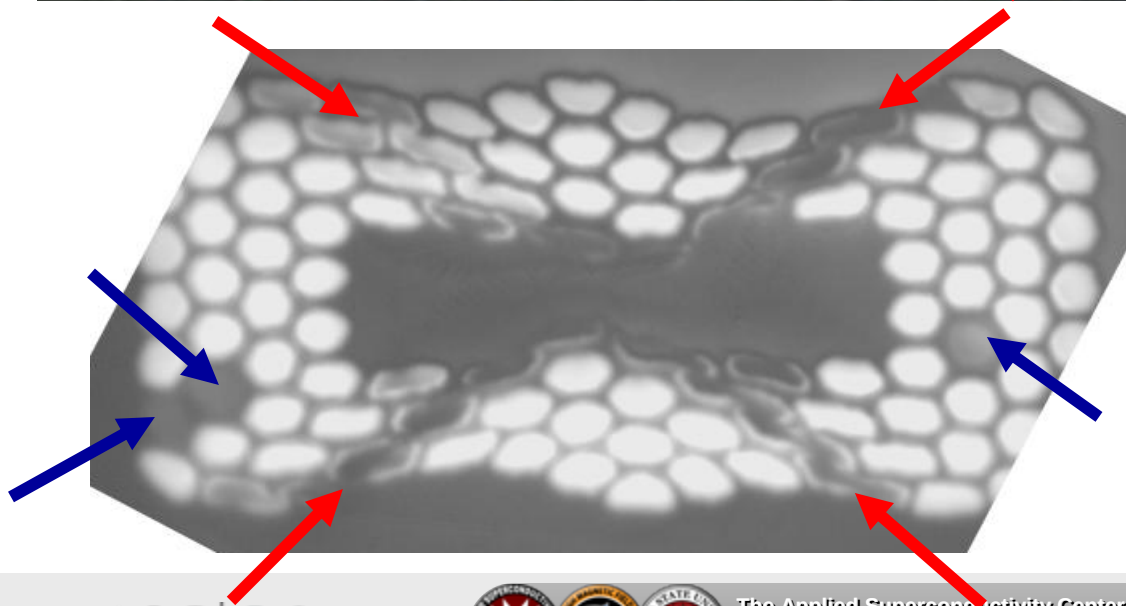
RRP Rolled to 0.6 mm: Color & MO: T=12 K, H=0



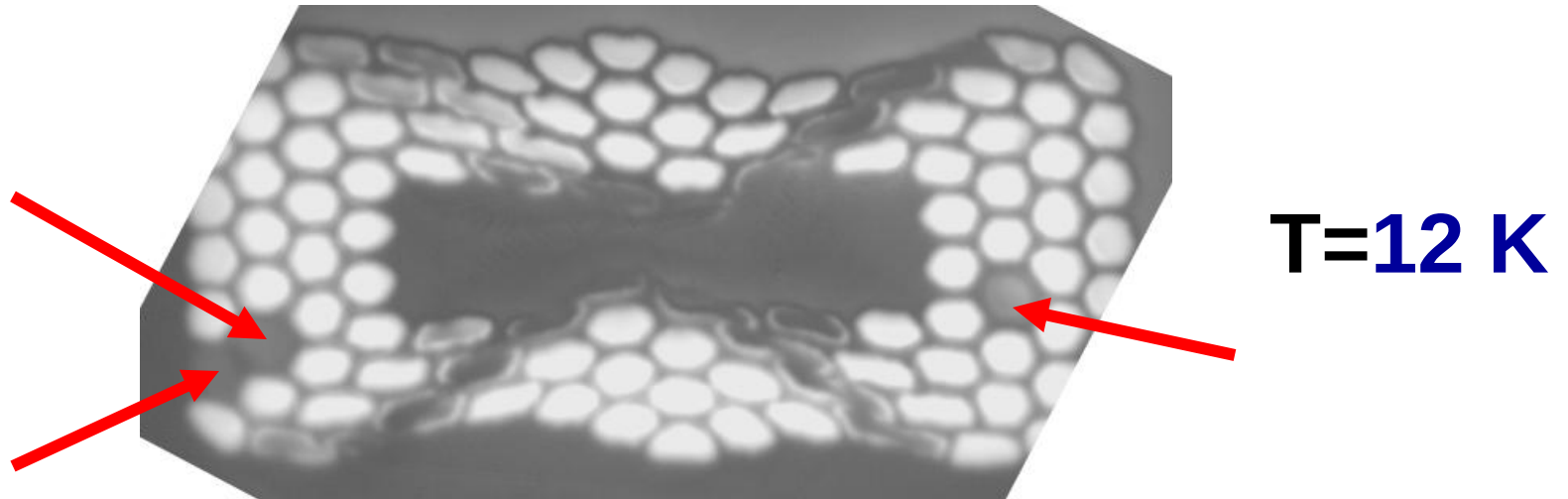
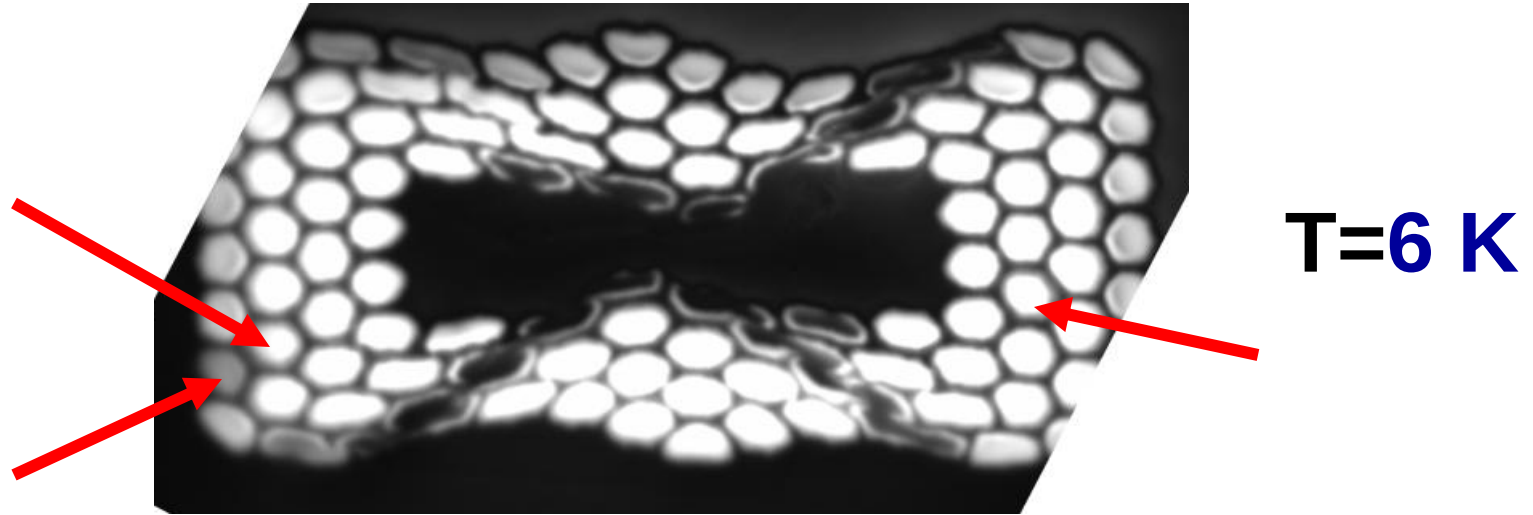
**Broken
subelements
along shear
planes**

**Low T_c
subelements
with partially
reacted Nb (Sn
loss from sub-
elements)**

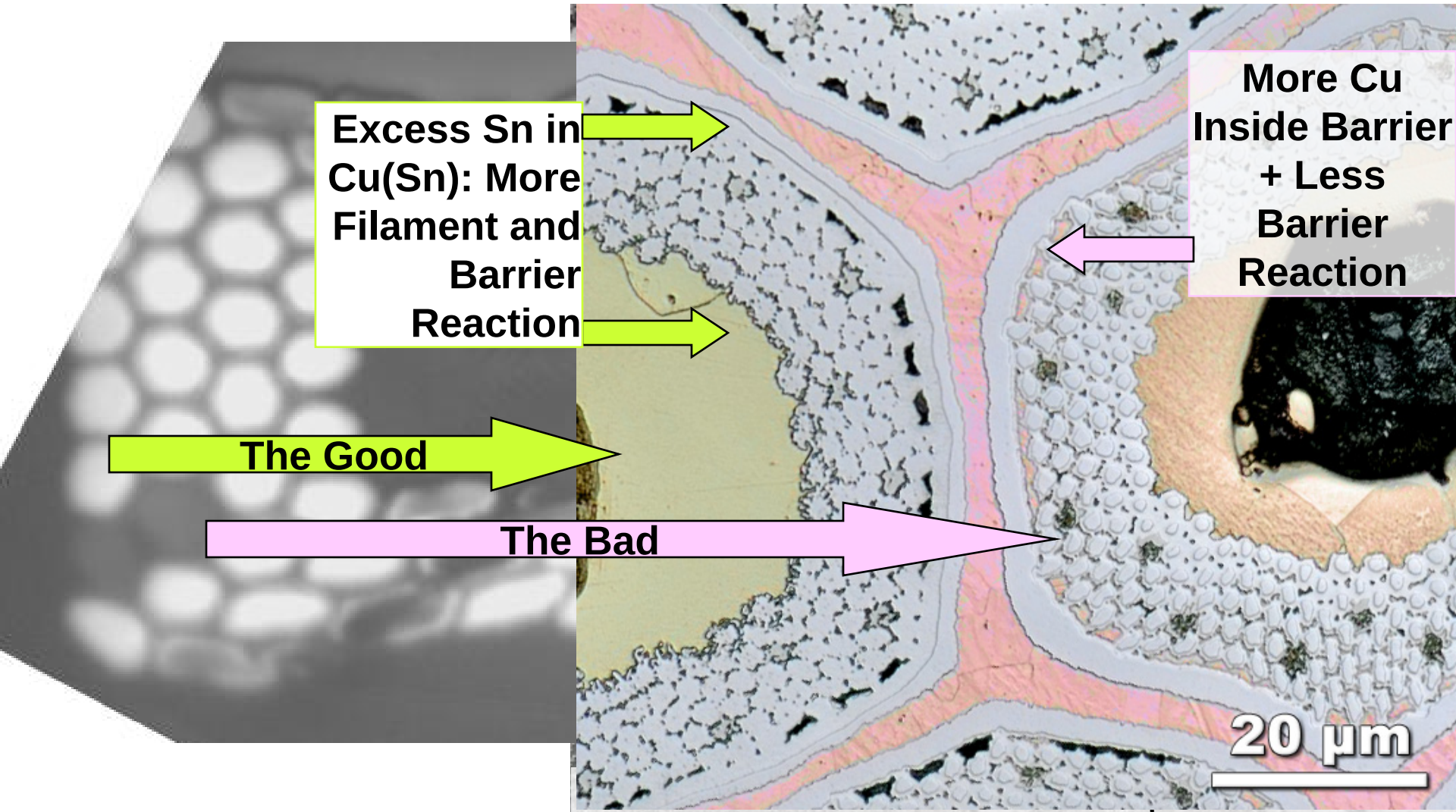
**MO shows two
types of defects
in RRP1**



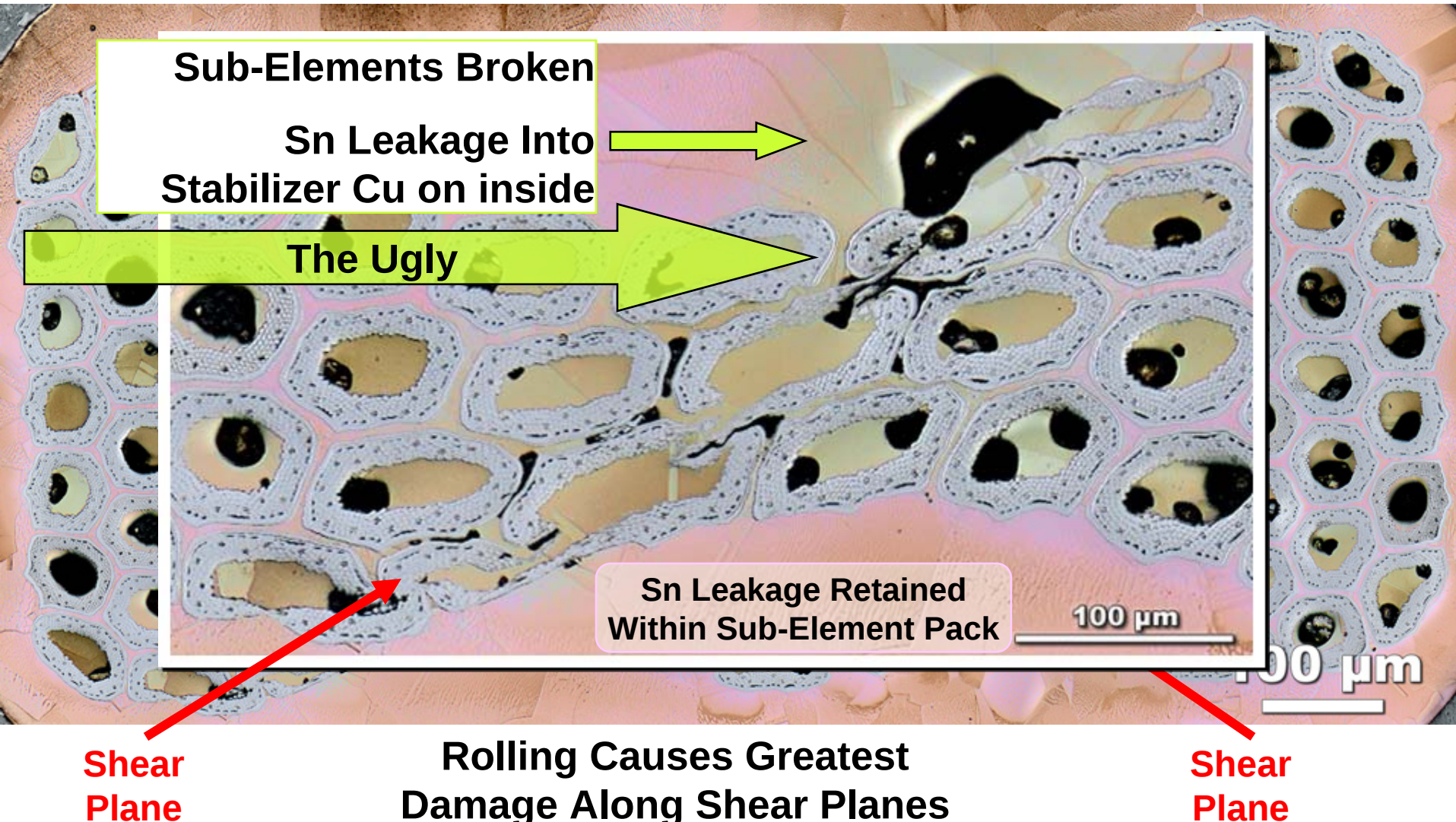
MO Shows T_c Reduction in Sub-Elements that are **Not** Damaged in this Cross-Section



Unreacted Nb Filament Cores In Low T_c Sub-Elements



Broken Sub-Elements In Shear Planes

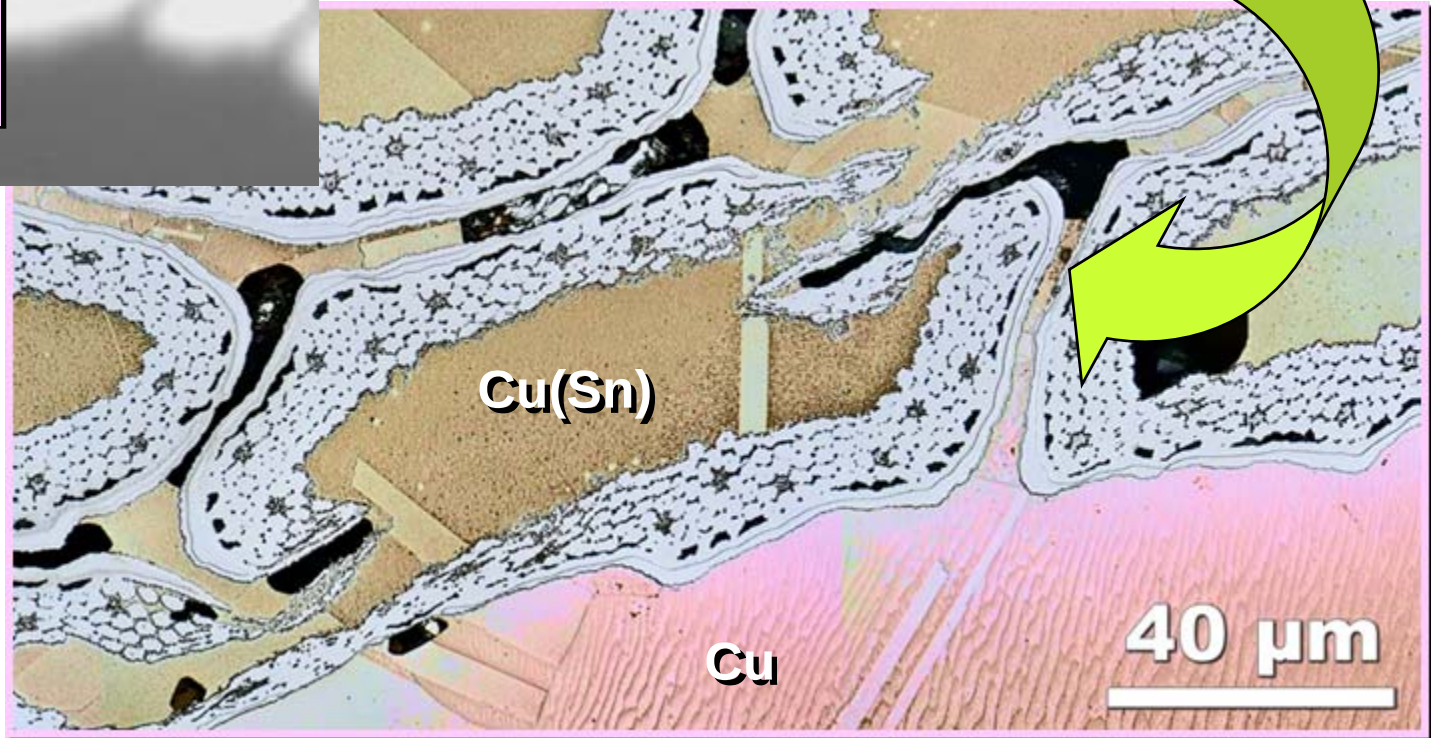


Sn Retained In Sub-Element Pack

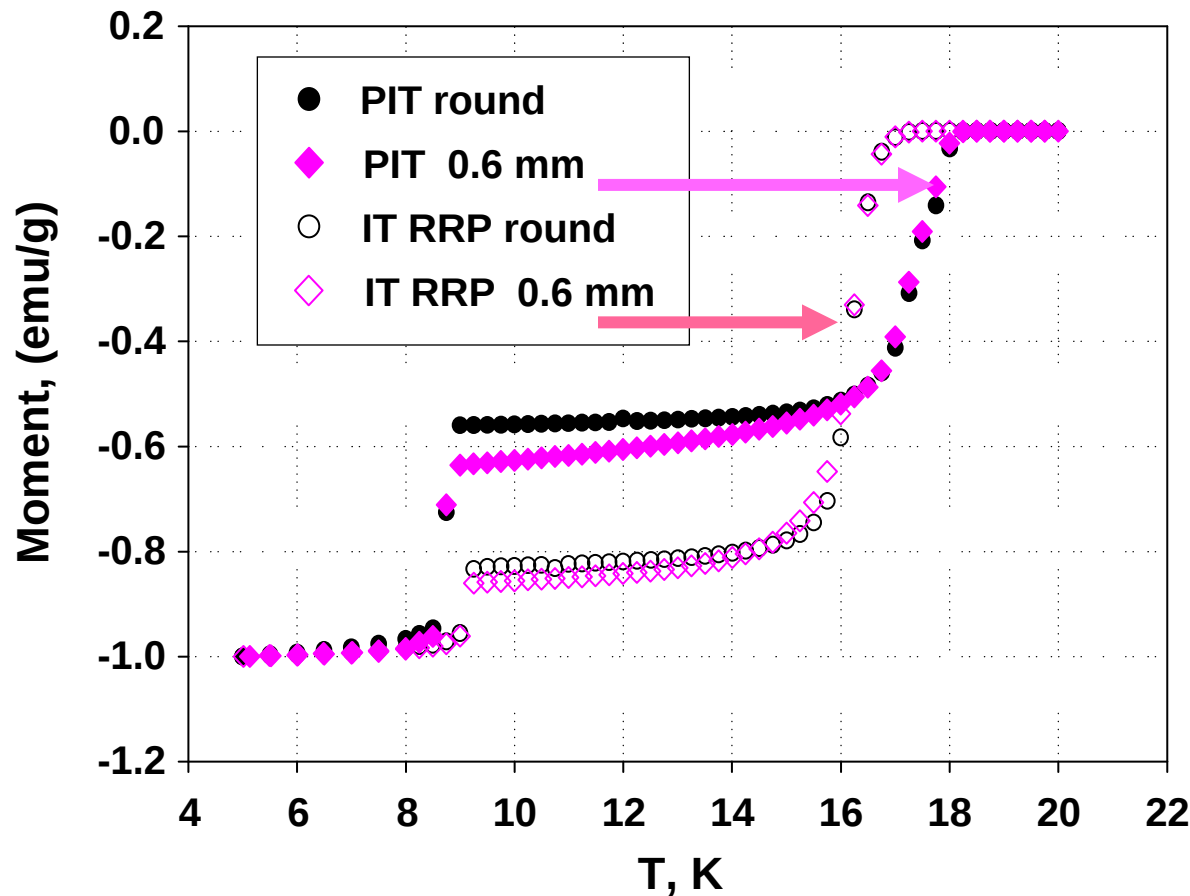
$T = 12 \text{ K}$

Restriction in Cu between sub-elements reduces Sn loss into outer Cu

Sufficient excess Sn retained to maintain superconductivity at 12 K



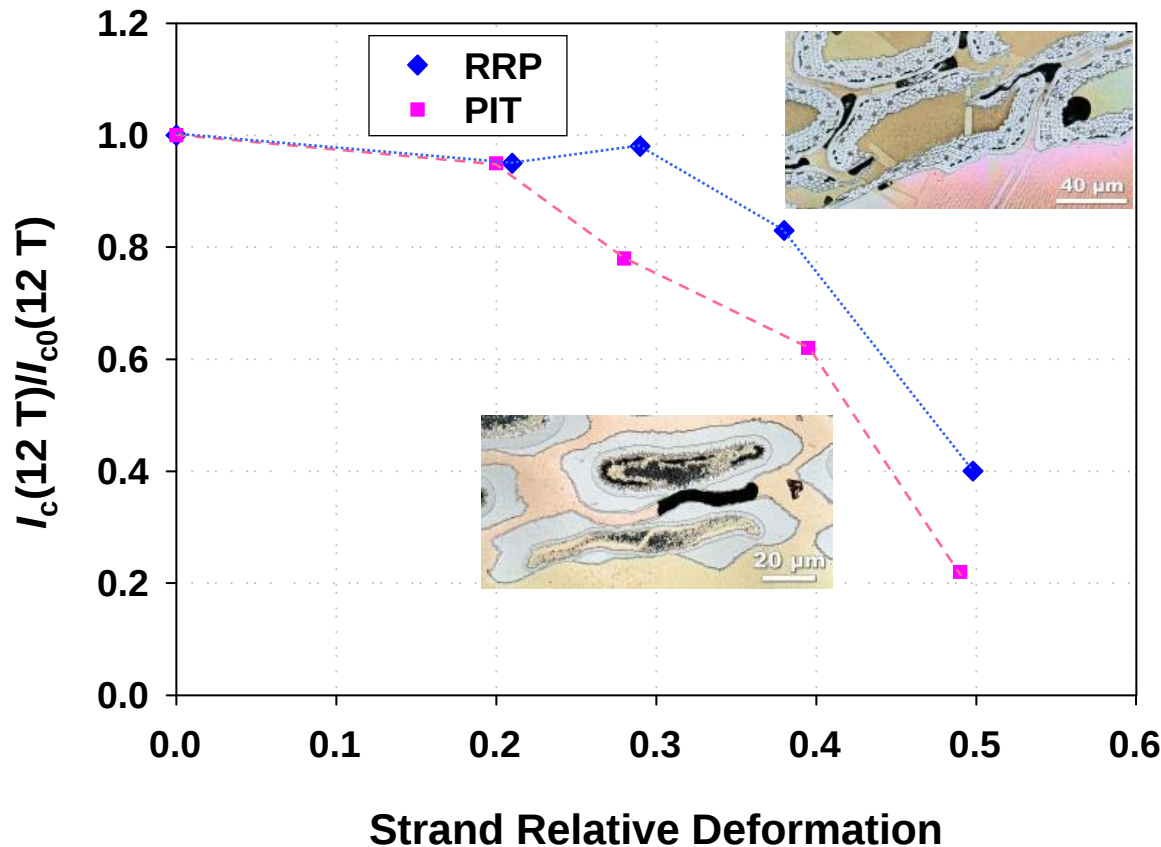
T_c Dependence vs. Rolling Deformation



The IT RRP strand uses Ti doping (from Nb-Ti rods) to enhance high field J_c and consequently has a lower T_c than the Nb(Ta)₃Sn of the PIT strand.

Deformation of the strands produces some broadening at the low end of the A15 transition due to lower-Sn A15 being formed.

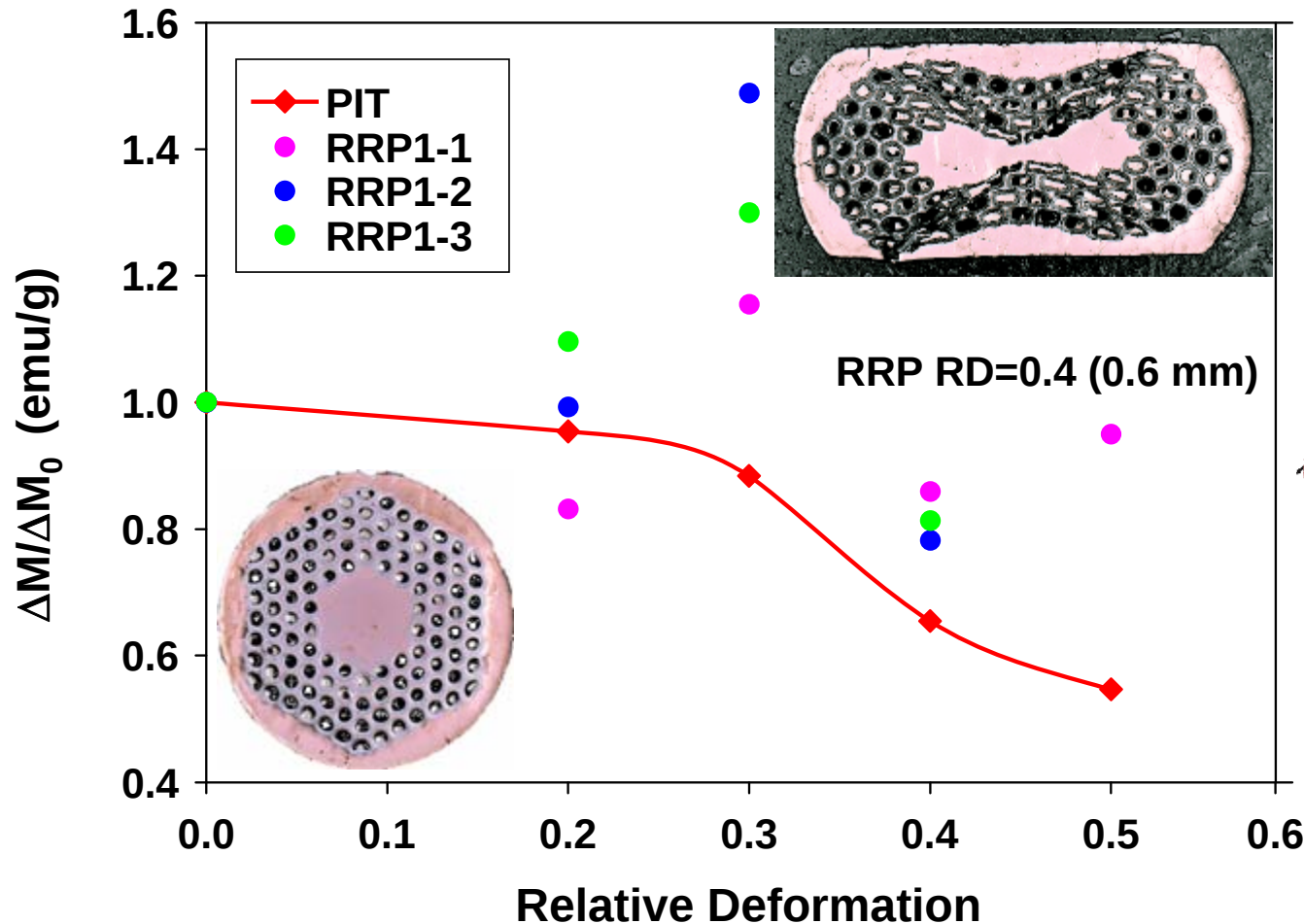
Transport I_c Dependence at 12 T (4.2 K)



Measurement at FNAL

- Both Strand Designs maintain J_c out to 80% of original diameter.
- PIT I_c drops faster after that perhaps because the short HT used for PIT strands (47-64 hr) means that loss of the high Sn is more catastrophic to A15 volume.

Magnetic Moment at T=12 K and H=2 T



Three different 5 mm RRP strand sets measured.

12 mm twist pitch applies different geometric distortions

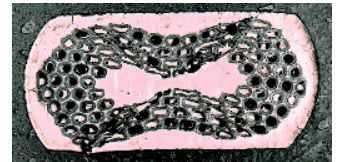
Magnetization measurements show considerable scatter in measurements for the deformed RRP strands

Conclusions

RRP and PIT behave differently under rolling deformation used to simulate cabling damage.

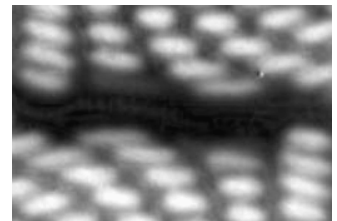
RRP:

-  In high-Sn RRP, Sn leaked from broken sub-elements but trapped in the sub-element pack can still be used locally to produce current carrying Nb_3Sn (indicated by MO).
-  Deformation inhomogeneous along length because of twisting.



PIT:

1. Sn lost from filament breakage in PIT strands results in critical loss of A15 layer.



Acknowledgments: Van Griffin, Jianyi Jiang, and Natanette Craig for measurement assistance